# Multianvil High-Pressure Synthesis of $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ : The First Oxoborate with Edge-Sharing $\mathrm{BO}_{4}$ Tetrahedra 

Hubert Huppertz* and Benjamin von der Eltz<br>Department of Chemistry, Ludwig-Maximilians-University of Munich, Butenandtstrasse 5-13, 81377 München, Germany<br>Received December 6, 2001

In the past decades oxoborates have developed into an important field of mineralogical and industrial interest. In particular the properties of the rare earth oxoborates, which exhibit high thermal stabilities and luminescence, have led to a variety of studies concerning high-temperature applications, optical glasses, phosphors, and NLO-materials. ${ }^{1}$

The structural chemistry of oxoborates is analoguous to that of silicates. The main difference is the possibility of boron to coordinate with oxygen not only in four-fold coordination (tetrahedral) but also in three-fold coordination (triangular). Until now it is not clear which one of the two coordination spheres is preferred under ambient conditions. For example, $\alpha$-metaboric acid $\left(\alpha-\mathrm{HBO}_{2}\right)$ exhibits only trigonal $\mathrm{BO}_{3}$ groups, $\beta$-metaboric acid $\left(\beta-\mathrm{HBO}_{2}\right)$ has trigonal and tetrahedral coordinations, and $\gamma$-metaboric acid ( $\gamma$ $\mathrm{HBO}_{2}$ ) shows exclusively tetrahedral coordination of boron. ${ }^{2}$ In analogy to the silicates the $\mathrm{BO}_{3}$ and $\mathrm{BO}_{4}$ groups can occur isolated or condensed in the form of groups, rings, chains, layers, or networks. Over the past decade Burns et al. developed a comprehensive description based on fundamental building blocks "FBB" to have a clearer nomenclature for more complicated polyanions. ${ }^{3,4}$

Primary high-pressure investigations on the rare-earth oxoborates $R E \mathrm{BO}_{3}(R E=\mathrm{Pr}, \mathrm{Nd}, \mathrm{Sm}-\mathrm{Dy}, \mathrm{Yb})$ were performed by Meyer. ${ }^{5}$ In these phases all boron atoms were still three-fold coordinated. Corresponding to the pressure-coordination rule ${ }^{6}$ an increase of pressure can lead to a higher coordination number of the boron atom, e.g. in $\mathrm{B}_{2} \mathrm{O}_{3}-\mathrm{I}$ all boron atoms have the coordination number three, whereas in the high-pressure modification $\mathrm{B}_{2} \mathrm{O}_{3}-\mathrm{II}$ all boron atoms are coordinated tetrahedrally. ${ }^{7}$ As a common feature of all nearly 500 structurally characterized oxoborates it can be pointed out that the $\mathrm{BO}_{3}$ triangles and the $\mathrm{BO}_{4}$ tetrahedra link to each other only via common corners (oxygen atoms) and not via edges or faces, to form rigid boron oxygen groups, that constitute the fundamental building blocks (FBB) of the structure. These distinguishing features were already postulated 1969 by Edwards and Ross. ${ }^{3,4,8}$

In this study we describe the synthesis and structure of a new oxoborate $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$, which differs structurally from all known oxoborates in a remarkable way: for the first time we observe that the exclusively appearing $\mathrm{BO}_{4}$ tetrahedra can link together not only via common corners but also via common edges. Therefore, one of the main postulates of oxoborate chemistry, that $\mathrm{BO}_{3}$ and $\mathrm{BO}_{4}$ polyhedra only share corners, is no longer valid.
$\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ was synthesized starting from stoichiometric amounts of boron oxide $\mathrm{B}_{2} \mathrm{O}_{3}$ and dysprosium oxide $\mathrm{Dy}_{2} \mathrm{O}_{3}$ in a multianvil apparatus under a pressure of 8 GPa and a temperature of $1000{ }^{\circ} \mathrm{C}$ (Scheme 1). We used a modified Walker module and a 1000 -t press. ${ }^{9}$ As a pressure medium precast MgO octahedra (Ceramic Substrates \& Components Ltd., Isle of Wight) with an edge length

[^0]

Figure 1. Crystal structure of $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$, viewed along [010]. The structure is built up from corrugated layers of corner-sharing (light) and edge-sharing (dark) $\mathrm{BO}_{4}$ tetrahedra. The $\mathrm{Dy}^{3+}$ ions are positioned between the layers.
of 18 mm were used. Eight tungsten carbide cubes with a truncation of 11 mm separated by pyrophyllite gaskets compressed the octahedra. ( $18 / 11$ assembly in conventionell terminology). The mixture was filled in a cylindrical boron nitride cylinder that was sealed by a boron nitride plate. The sample cylinder was placed at the center of a cylindrical resistance heater (graphite) that had a variable (stepped) wall thickness to minimize the thermal gradient along the sample. MgO rods filled the space on the top and bottom of the sample. A cylindrical zirconia sleeve surrounding the furnace provided thermal insulation. The assembly was positioned inside the octahedra and contacted with a molybdenum ring at the top and a molybdenum plate at the bottom. The experimental temperature was monitored using a $\mathrm{Pt} / \mathrm{Pt}_{87} \mathrm{Rh}_{13}$ thermocouple that was inserted axially into the octahedral assembly, with the hot junction in contact with the boron nitride cylinder.

## Scheme 1

$$
2 \mathrm{Dy}_{2} \mathrm{O}_{3}+3 \mathrm{~B}_{2} \mathrm{O}_{3} \xrightarrow[1000^{\circ} \mathrm{C}]{8.0 \mathrm{GPa}} \mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}
$$

$\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ synthesized under these conditions is a colorless, nearly phase-pure, coarse crystalline solid (yield: 40 mg per run). Elemental analyses of dysprosium and boron by ICP on a VarianVista spectrometer revealed $68.7 \mathrm{wt} \%$ Dy and $6.6 \mathrm{wt} \%$ B (calcd: $68.1 \% \mathrm{Dy}, 6.8 \% \mathrm{~B})$. The absence of hydrogen $(\mathrm{OH})$ was checked by IR spectroscopy. X-ray structure analysis on single crystals showed that the structure of $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ is built up from corrugated layers of linked $\mathrm{BO}_{4}$ tetrahedra (Figure 1). ${ }^{10}$ The $\mathrm{Dy}^{3+}$ ions are positioned between the layers.
$\mathrm{BO}_{4}$ tetrahedra inside the layers are linked via common corners as well as common edges (Figure 2). In detail the edge-sharing tetrahedra (dark polyhedra) form six-membered rings with two additional corner-sharing $\mathrm{BO}_{4}$ tetrahedra (light polyhedra). The linkage of these rings by further corner-sharing $\mathrm{BO}_{4}$ tetrahedra leads to 10 -membered rings forming corrugated layers.


Figure 2. View on a corrugated layer of $\mathrm{BO}_{4}$ tetrahedra in $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$. The linkage of the tetrahedra to rings is via common corners as well as via common edges.


Figure 3. Interatomic distances in pm and angles in deg inside the edgesharing $\mathrm{BO}_{4}$ tetrahedra of $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$.

Ionic oxotetrahedra do not normally link together via common edges. In 1965 a work ${ }^{11}$ about "fibrous $\mathrm{SiO}_{2}$ " was published describing edge-sharing $\mathrm{SiO}_{4}$ tetrahedra. A verification of these results was not possible. In the higher homologues of the boron and silicon chalcogenides $\left(\mathrm{B}_{2} \mathrm{~S}_{3},{ }^{12-14} \mathrm{SiS}_{2},{ }^{15} \mathrm{SiSe}_{2},{ }^{15} \mathrm{SiTe}_{2}{ }^{16}\right)$ the formation of $\mathrm{B}_{2} \mathrm{~S}_{2}, \mathrm{Si}_{2} \mathrm{~S}_{2}, \mathrm{Si}_{2} \mathrm{Se}_{2}$, and $\mathrm{Si}_{2} \mathrm{Te}_{2}$ rings is not uncommon. This is due to the lower ionicity and the remarkable larger distances inside the four-membered rings. Because of the fact, that oxosilicates do not exhibit edge-sharing $\mathrm{SiO}_{4}$ tetrahedra, it is much more remarkable for boron as an element of the first short period to form edge-sharing $\mathrm{BO}_{4}$ tetrahedra under these reaction conditions. One of the main reasons is the lower formal charge of boron $(+3)$ in contrast to silicon $(+4)$. Until now only five crystal structures containing four-membered $\mathrm{B}_{2} \mathrm{O}_{2}$ rings have been reported in molecular chemistry, but none of these examples shows boron exclusively coordinated by oxygen. ${ }^{17,18}$

The $\mathrm{B}-\mathrm{O}$ bond lengths inside the corner-sharing $\mathrm{BO}_{4}$ tetrahedra of $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ vary between 143 and 152 pm . Inside the $\mathrm{B}_{2} \mathrm{O}_{2}$ ring of the edge-sharing tetrahedra the $\mathrm{B}-\mathrm{O}$ distances are slightly longer (150.7(5) and 153.3(5) pm) (Figure 3). The average $\mathrm{B}-\mathrm{O}$ distance of 147.9 pm in $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ is in good agreement with the average $\mathrm{B}-\mathrm{O}$ bond length of 147.6 pm in $\mathrm{BO}_{4}$ tetrahedra. ${ }^{19}$ The transannular $\mathrm{B} \cdots \mathrm{B}$ distance $(207.2(8) \mathrm{pm})$ in the $\mathrm{B}_{2} \mathrm{O}_{2}$ ring is markedly shorter than in $\mathrm{B}_{2} \mathrm{~S}_{3}(224 \mathrm{pm}) .{ }^{13}$ Both crystallographically different $\mathrm{Dy}^{3+}$ ions are coordinated from eight oxygen atoms in the range 224265 pm . A classification of $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ with the help of the " FBB "concept after Burns et al. ${ }^{4}$ will follow when the concept is extended by a descriptor for edge-sharing tetrahedra.

Raman- and IR-spectroscopic investigations were performed. A clear classification of the vibrational bands is an issue of great complexity because comparative data for edge-sharing $\mathrm{BO}_{4}$ tetrahedra are not available. To investigate the metastable character of the high-pressure phase $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ temperature-dependent X-ray powder diffraction measurements were performed on a STOE powder diffractometer Stadi P . Above $700{ }^{\circ} \mathrm{C} \mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ decomposes into the hexagonal phase $\pi$ - $\mathrm{DyBO}_{3}$ (room-temperature phase),
$\mu$ - $\mathrm{DyBO}_{3}$ (high-temperature phase), and $\mathrm{B}_{2} \mathrm{O}_{3}$. Further raising the temperature to $1100{ }^{\circ} \mathrm{C}$ leads to a complete transformation of $\pi$ - $\mathrm{DyBO}_{3}$ into the high-temperature phase. Experiments to synthesize $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}$ under normal pressure resulted in a mixture of $\pi$ - $\mathrm{DyBO}_{3}$ and $\mathrm{B}_{2} \mathrm{O}_{3}$ as a byproduct.

Acknowledgment. We thank Professor Dr. W. Schnick for his continuous support. We thank Dr. P. Ulmer (ETH Zürich), Dr. D. J. Frost, and Professor Dr. D. C. Rubie (Bayerisches Geoinstitut Bayreuth) for the help adapting the multianvil technique. We also thank Dr. H. Piotrowski (Ludwig-Maximilians-Universität München) for collecting single-crystal data.

Supporting Information Available: Crystallographic data and further details (CIF). This material is available free of charge via the Internet at http://pubs.acs.org or from the Fachinformationzentrum Karlsruhe, D-76344 Eggenstein-Leopoldshafen (Germany), e-mail: crysdata@fiz-karlsruhe.de, by quoting the registry number CSD-412041.

## References

(1) (a) Gmelin Handbook of Inorganic and Organometallic Chemistry C11b, 8th ed; Nohl, U., Olbrich, G., Eds.; Springer: Berlin, 1991. (b) Becker, P. Adv. Mater. 1998, 10, 979-992.
(2) (a) $\alpha-\mathrm{HBO}_{2}$ : Peters, C. R.; Milberg, M. E. Acta Crystallogr. 1964, 17, 229-234. (b) $\beta$ - $\mathrm{HBO}_{2}$ : Zachariasen, W. H. Acta Crystallogr. 1963, 16, 385-389. (c) $\gamma-\mathrm{HBO}_{2}$ : Zachariasen, W. H. Acta Crystallogr. 1963, 16, 380-384.
(3) (a) Edwards, J. O.; Ross, V. F. J. Inorg. Nucl. Chem. 1960, 15, 329337. (b) Christ, C. L. Am. Mineral. 1960, 45, 334-340. (c) Tennyson, C. Fortschr. Mineral. 1963, 41, 64-91. (d) Ross, V. F.; Edwards, J. O. The Structural Chemistry of the Borates. In The Chemistry of Boron and its Compounds; Wiley: New York, 1967. (e) Heller, G. Fortschr. Chem. Forsch. 1970, 15, 206-280. (f) Christ, C. L.; Clark, J. R. Phys. Chem. Miner. 1977, 2, 59-87.
(4) (a) Burns, P. C.; Grice, J. D.; Hawthorne, F. C. Can. Mineral. 1995, 33, 1131-1151. (b) Grice, J. D.; Burns, P. C.; Hawthorne, F. C. Can. Mineral. 1999, 37, 731-762.
(5) (a) Meyer, H. J. Naturwissenschaften 1969, 56, 458-459. (b) Meyer, H. J.; Skokan, A. Naturwissenschaften 1971, 58, 566. (c) Meyer, H. J. Naturwissenschaften 1972, 59, 215.
(6) Neuhaus, A. Chimia 1964, 18, 93-103.
(7) (a) Prewitt, C. T.; Shannon, R. D. Acta Crystallogr. 1968, B24, 869874. (b) Gurr, G. E.; Montgomery, P. W.; Knutson, C. D.; Gorres, B. T. Acta Crystallogr. 1970, B26, 906-915.
(8) Filatov, S. K.; Bubnova, R. S. Phys. Chem. Glasses 2000, 41 (5), 216224.
(9) (a) Huppertz, H. Z. Naturforsch. 2001, 56b, 697-703. (b) Walker, D.; Carpenter, M. A.; Hitch, C. M. Am. Mineral. 1990, 75, 1020-1028. (c) Walker, D. Am. Mineral. 1991, 76, 1092-1100. (d) Rubie, D. C. Phase Transitions 1999, 68, 431-451.
(10) Crystal structure analysis of $\mathrm{Dy}_{4} \mathrm{~B}_{6} \mathrm{O}_{15}: \mathrm{MW}=954.86 \mathrm{~g} / \mathrm{mol}$, colorless block, $0.03 \times 0.05 \times 0.06 \mathrm{~mm}^{3}$, monoclinic, space group: $C 2 / c, a=$ 1169.4(1) pm, $b=437.8(1) \mathrm{pm}, c=1893.0(1) \mathrm{pm}, \beta=96.66(1)^{\circ}, V=$ $0.960(2) \mathrm{nm}^{3}, Z=4, \rho_{\text {calc }}=6.589 \mathrm{~g} / \mathrm{cm}^{3}$, Enraf-Nonius Kappa CCD with rotating anode, Mo K $\alpha$-radiation $(\lambda=71.073 \mathrm{pm})$, graphite-monochromator, $F(000)=1656, \mu=30.8 \mathrm{~mm}^{-1}, T=293(2) \mathrm{K}, \varphi / \omega$-scan, 11351 measured reflections in the range $7^{\circ}<2 \theta<60^{\circ}$, 1405 unique reflections with $I \geq 0 \sigma(I)$, numerical absorption correction (HABITUS), transmissions ratio: $2.25(\mathrm{max} / \mathrm{min}), R_{\text {int }}=0.0439$; the crystal structure was solved by direct methods (SHELXS-97) and anisotropically refined by a least squares procedure against $F^{2}$ with all data, 115 refined parameters, R1 $=0.0155$ and wR2 $=0.0349$ for $F_{0}{ }^{2} \geq 2 \sigma\left(F_{0}{ }^{2}\right) ; \mathrm{R} 1=0.0173$ and $\mathrm{wR} 2=0.0356$ for all $F_{\mathrm{o}}{ }^{2}, \mathrm{GOF}=1.098$.
(11) Weiss, A.; Weiss, A. Z. Anorg. Allg. Chem. 1954, 276, 95-112.
(12) Conrad, O.; Jansen, C.; Krebs, B. Angew. Chem. 1998, 110, 3396-3407; Angew. Chem., Int. Ed. 1998, 37, 3208-3218.
(13) Diercks, H.; Krebs, B. Angew. Chem. 1977, 89, 327-328; Angew. Chem., Int. Ed. Engl. 1977, 16, 313-314. In $\mathrm{B}_{2} \mathrm{~S}_{3}$ there is a $\mathrm{B}_{2} \mathrm{~S}_{2}$ ring inside of ${ }_{14} \mathrm{~B}_{2} \mathrm{~S}_{4}{ }^{2-}$ unit. Edge-sharing $\mathrm{BS}_{4}$ tetrahedra can be found in $\mathrm{TlBS}_{2}$ (cf. ref 14).
(14) Krebs, B.; Hamann, W. Z. Kristallogr. 1983, 162, 149-150.
(15) Peters, J.; Krebs, B. Acta Crystallogr. 1982, B38, 1270-1272.
(16) Weiss, A.; Weiss, A. Z. Naturforsch. 1953, 8b, 104.
(17) (a) Heller, G.; Giebelhausen, A. Z. Anorg. Allg. Chem. 1980, 460, 228234. (b) Cayton, R. H.; Chisholm, M. H.; Davidson, E. R.; DiStasi, V. F.; Du, P.; Huffmann, J. C. Inorg. Chem. 1991, 30, 1020-1024.
(18) Burke, J. M.; Fox, M. A.; Goeta, A. E.; Hughes, A. K.; Marder, T. B. Chem. Commun. 2000, 2217-2218.
(19) Hawthorne, F. C.; Burns, P. C.; Grice, J. D. Reviews in Mineralogy. In Boron: Mineralogy, Petrology, and Geochemistry; Mineralogical Society of America: Washington DC, 1996; Chapter 2, Vol. 33.
JA017691Z


[^0]:    * To whom correspondence should be addressed. E-mail: huh@cup.unimuenchen.de.

